

## Semiconductor Quantum Dots



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"Semiconductor Physics and Nanoscience" St. Petersburg – JASS 2005





- Introduction
- Fabrication Experiments Applications
  - Porous Silicon
  - II-VI Quantum Dots
  - III-V Quantum Dots
    - Cleaved Edge Overgrowth (CEO)
    - Self Assembling Quantum Dots
    - Electronic Structure

### Introduction Low Dimensional Systems



 Motion of electron in conduction band is described by the effective mass concept

$$E = \frac{p^2}{2m^*}$$

• Dispersion relation with  $p = \hbar k$ 

$$\Rightarrow E(k) = \frac{\hbar^2 k^2}{2m^*}$$

 In low dimensional systems the carrier motion is quantized in one or more spatial directions



## Inroduction Density of States – 3D



• Wave function in 3D box of volume  $\Omega = L_x L_y L_z$ 

$$\Phi_{lmn}(\mathbf{R}) = \frac{1}{\sqrt{\Omega}} \exp(i\mathbf{K} \cdot \mathbf{R}) \qquad \mathbf{K} = \left(\frac{2\pi l}{L_x}, \frac{2\pi m}{L_y}, \frac{2\pi n}{L_z}\right)$$

Density of states / per unit volume

$$N_{3D}(\mathbf{K}) = \frac{2\Omega}{(2\pi)^3} \frac{4}{3} \mathbf{K}^3 \pi \qquad n_{3D}(\mathbf{K}) = \frac{1}{3\pi^2} \mathbf{K}^3$$

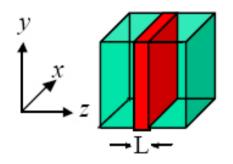
Density of states in Energy

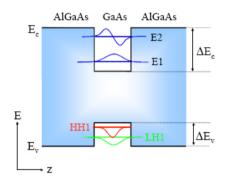
$$D_{3D}(E) = \frac{d}{dE} n_{3D}(\mathbf{K}) = \frac{1}{2\pi^2} \left(\frac{2m^*}{\hbar^2}\right)^{\frac{3}{2}} \sqrt{E - E_g}$$





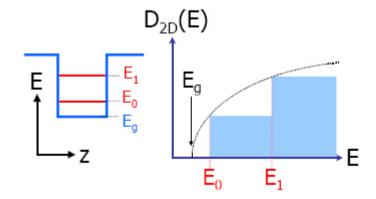
• For example  $GaAs/Al_xGa_{1-x}As(x<0.4)$  quantum well





Density of states

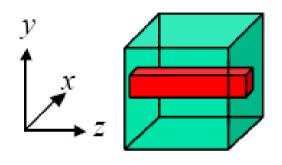
$$D_{2D}\left(E\right) = \frac{m^*}{\pi \hbar^2}$$

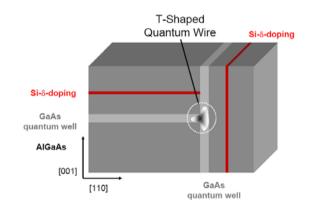






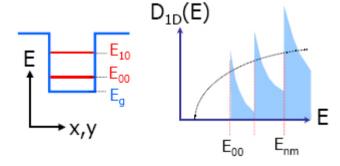
Quantum wire through cleaved edge overgrowth





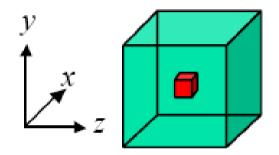
Density of states

$$D_{1D}(E) = \frac{\sqrt{2m^*}}{2\pi\hbar} \frac{1}{\sqrt{E - E_{nm}}}$$



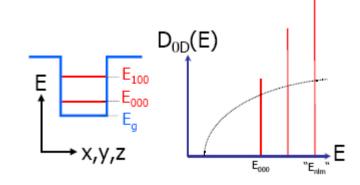


Kinetic quantization along x, y and z-direction



- Energy spectrum fully quantized
- Density of States

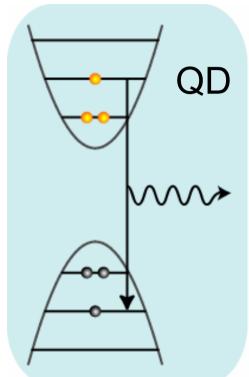
$$D_{0D}(E) = discrete$$



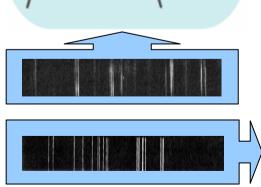


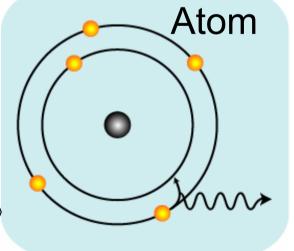
## **Introduction Quantum Dots as Artificial Atoms**





	Atom	QD	
Particle:	Electron	on Exciton	
Energy scale:	13eV – 100keV	Typ. 1 eV	
Length scale:	0.1 – 0.3 nm	Typ. 10 nm	
Potential:	V(r) ~ 1/r	Tunable	





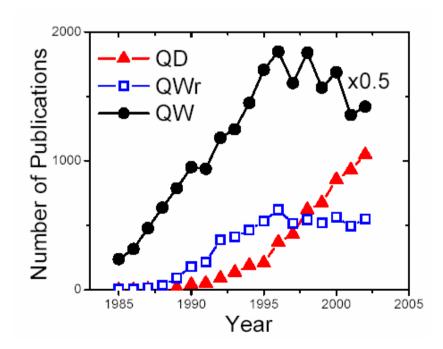


Tunable properties in QDs





- Applications
  - Lasers in visible and near infrared spectrum
  - Optical data storage
  - Optical detectors
  - Quantum information processing and cryptography
- Publications







- Size  $\Delta E > 3k_BT \sim 75meV$
- Crystal quality
- Uniformity
- Density
- Growth compatibility
- Confinement for electrons and/or holes
- Electrically active matrix material





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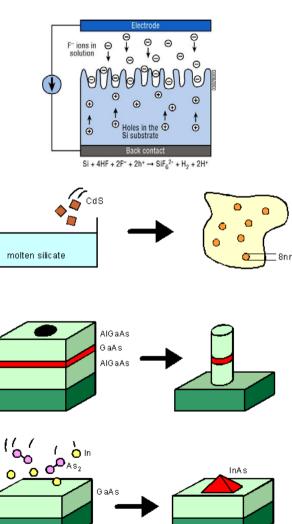


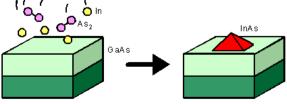
### Fabrication – Experiments – Applications Introduction



#### Several approaches:

- Porous Silicon
- Nanometer semiconductor inclusions in matrices
- Lithographic patterning of higher dimensional systems
- Strain driven selfassembly









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- C-Si: indirect bandgap → inefficient emitter even at 4K
- P-Si: emission efficiency up to 10% (optical excitation)
- Nanocrystals of different size and shape
- Structure of high complexity
- Confinement leads to bandgap widening and higher overlap of wavefunctions
- Easy fabrication of p-Si
- Pure Si optoelectronic devices possible

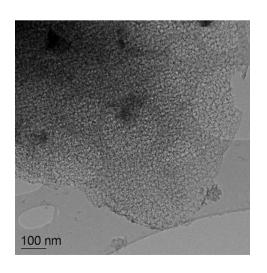


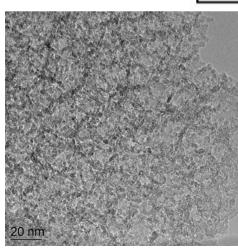


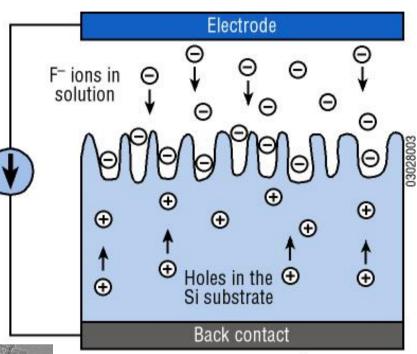
 Anodic biased c-Si in hydrofluoric acid (HF)

Structure depends on:

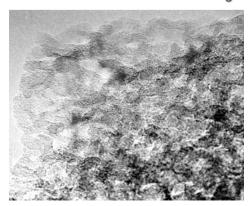
- Doping
- Etching conditions
- Illumination conditions







$$Si + 4HF + 2F^- + 2h^+ \rightarrow SiF_6^{2-} + H_2 + 2H^+$$

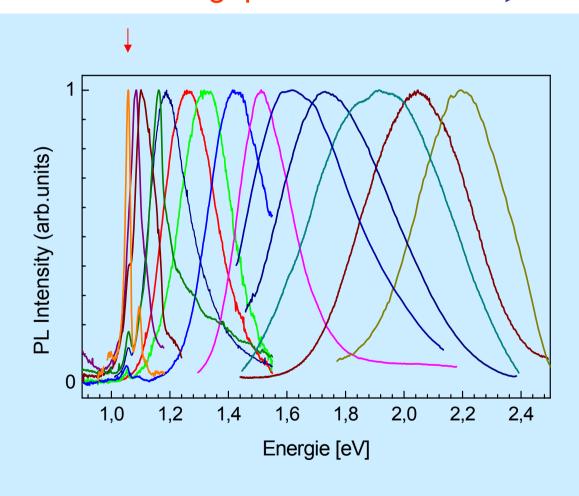






### Si bandgap

#### Smaller nc's size



Widely tunable emission band due to quantum size effect: all emission energies are available

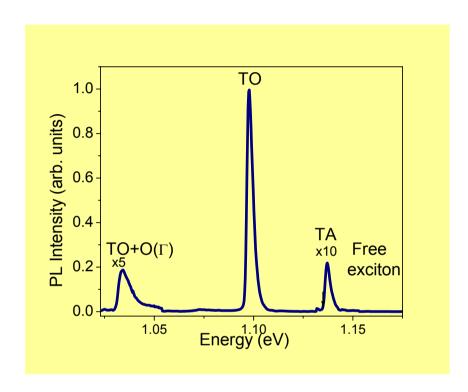
Broad spectrum

→ line narrowing

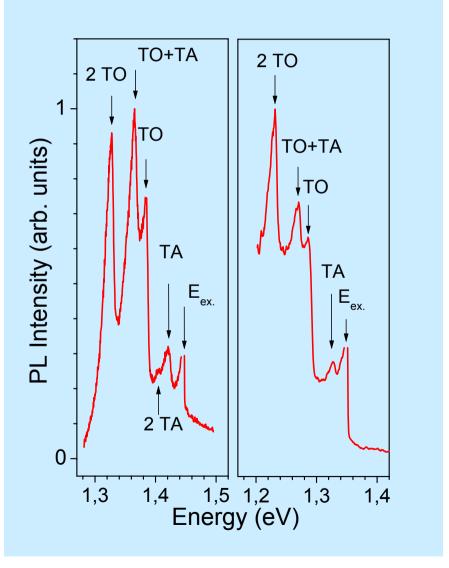


# Porous Silicon *k*-space





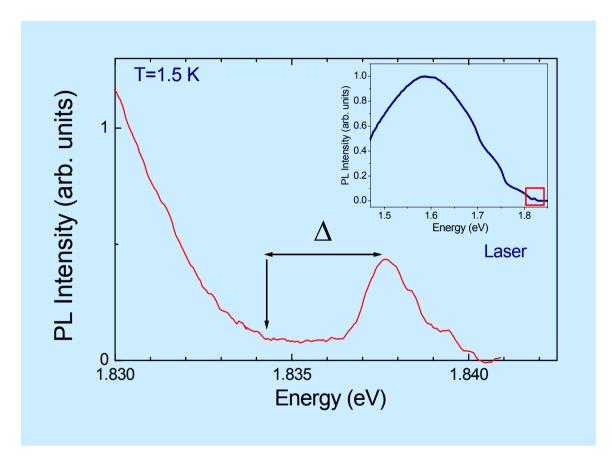
- P-Si has indirect nature
- k-conservation rule breaks down due to confinement

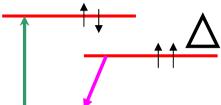




### Porous Silicon Electron-Hole Exchange Interaction







Absorption in a singlet state

After spin flip emission via triplet state

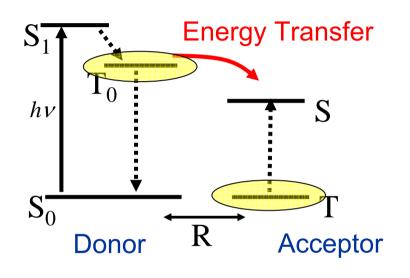
Electronic structure of excitons is very similar to dye molecules



## Porous Silicon Indirect Excitation: Photosensitization



Basic principle:



Energy transfer (dipole-dipole or direct electron exchange) is efficient if:

- photoexcited donor has long lifetime
- overlap of energy bands of D/A is good
- space separation of D/A is small

Silicon nanocrystals (almost ideal donor):

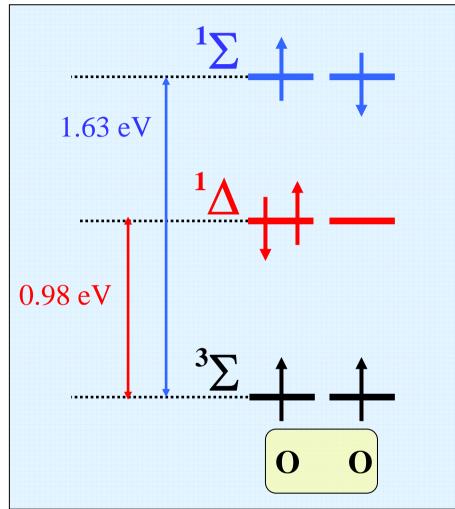
- ground state is triplet
- •long exciton lifetime (10<sup>-5</sup>-10<sup>-3</sup> s)
- wide emission band
- •huge internal surface area (10<sup>3</sup> m<sup>2</sup>/cm<sup>3</sup>)

Acceptor having triplet ground state?



### Porous Silicon Molecular Oxygen: Electronic Structure





ground gtate:

- spin triplet
- chemically inert
   (reaction S+T → S is forbidden)

#### excited states:

- spin singlet
- energy-rich
- high chemical reactivity
   (reaction S+S → S is allowed

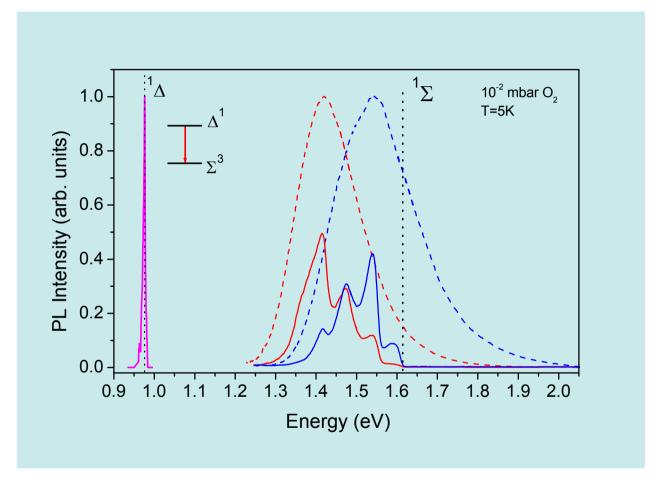
oxidation reactions in organic chemistry, biology, life science photodynamic cancer therapy oxygen-iodine laser

Optical excitation is impossible → Photosensitizer is required → Silicon nanocrystals



## Porous Silicon PL Quenching by Oxygen Molecules





- $\rightarrow$  Adsorption of O<sub>2</sub>: PL Quenching
- → low temperature: fine structure appears
  - ¹D→ ³S emission line of O<sub>2</sub>



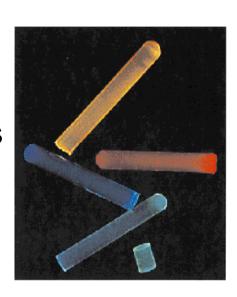


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- First hints of quantum dots: CdSe and CdS in silicate glasses (X-ray 1932)
- Since 1960s semiconductor doped glasses used as sharp-cut color filter
- Quantum dots in glassy matrices
- Ideal model for the study of basic concepts of 3D confinement in semiconductors
- Many different matrices: glasses, solutions, polymers, even cavities of zeoliths
- Many promising applications already on the way



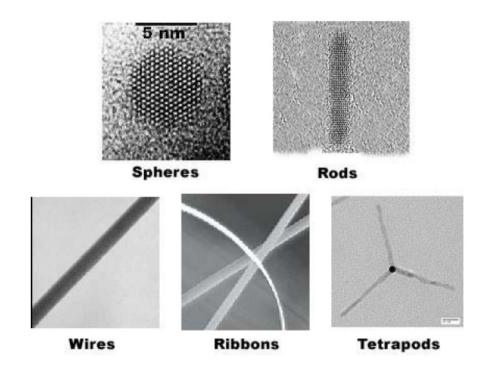




 Colloidal QDs can be further processed and incorporated in a variety of media



CdSe can be prepared in a wide range of shapes

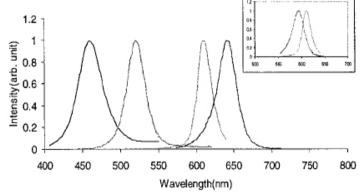


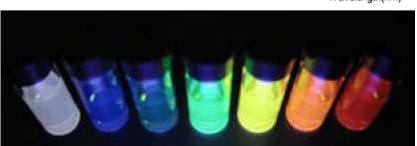


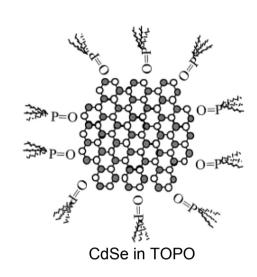
## II-VI Quantum Dots Growth of Nanocrystals

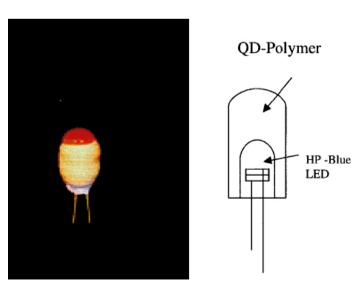


- In polymer composites
  - Nearly full color emitting LEDs
  - (CdSe)ZnS in PLMA (green red)
  - (CdS)ZnS in PLMA (violet blue)
  - (CdS)ZnS in PLMA for temperature measurements





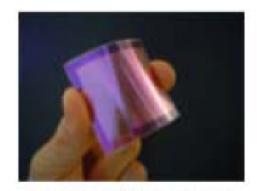








Coupled to bio-molecules → biological sensors



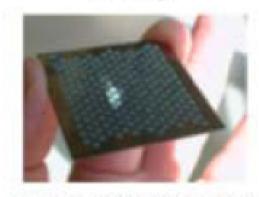
Conformal Solar Cells



Flexible Electronics



Memory

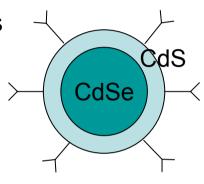


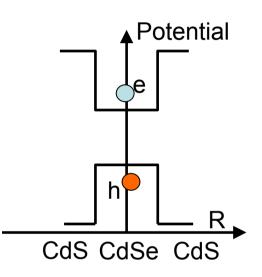
**Drug Discovery Substrates** 



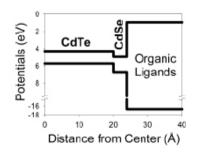


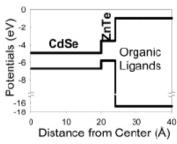
- Type-I core-shell structure (CdSe)CdS
  - Display devices and lasers

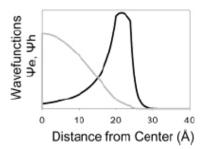


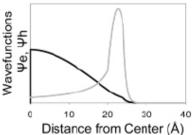


- Type-II (CdTe)CdSe and (CdSe)ZnTe
  - Esp. photovoltaic photoconducting devices
  - Energies smaller than bandgap of each material possible
  - Tunable bandgap low yield (<5%)</li>











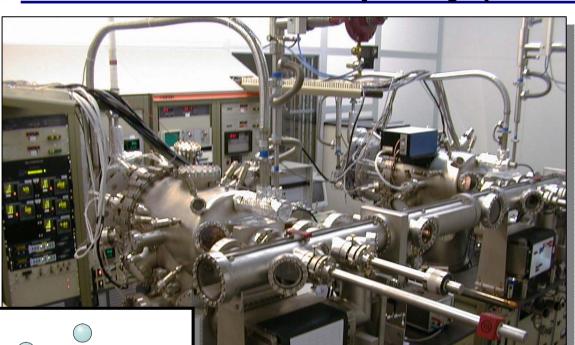


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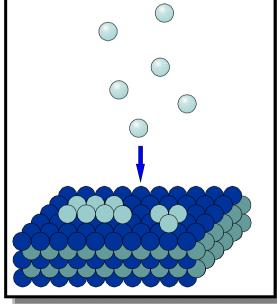


### CEO Molecular-Beam-Epitaxy (MBE)

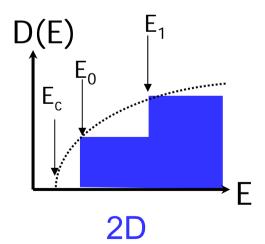




Quantum Well

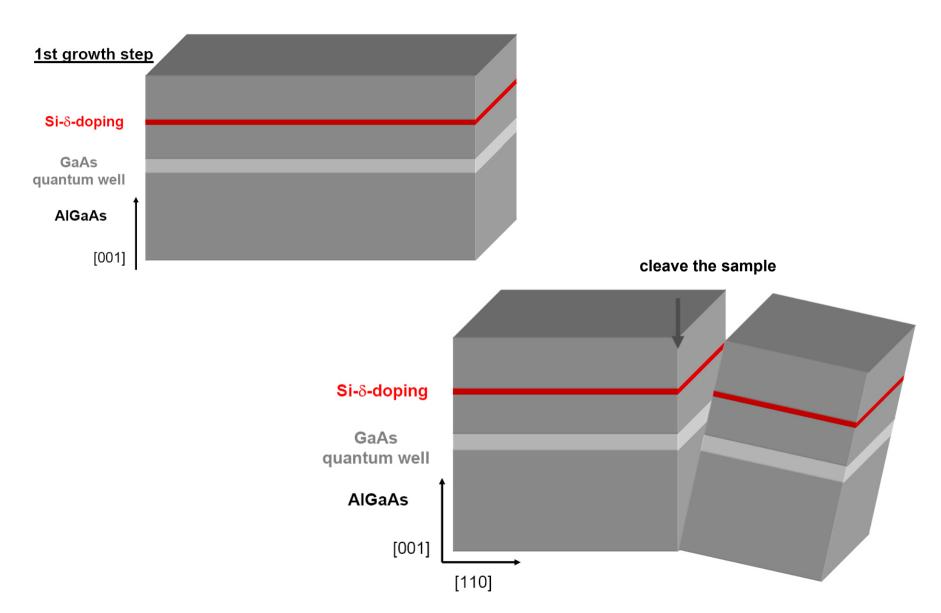


MBE→ Atomically precise deposition of layers with different composition and/or doping



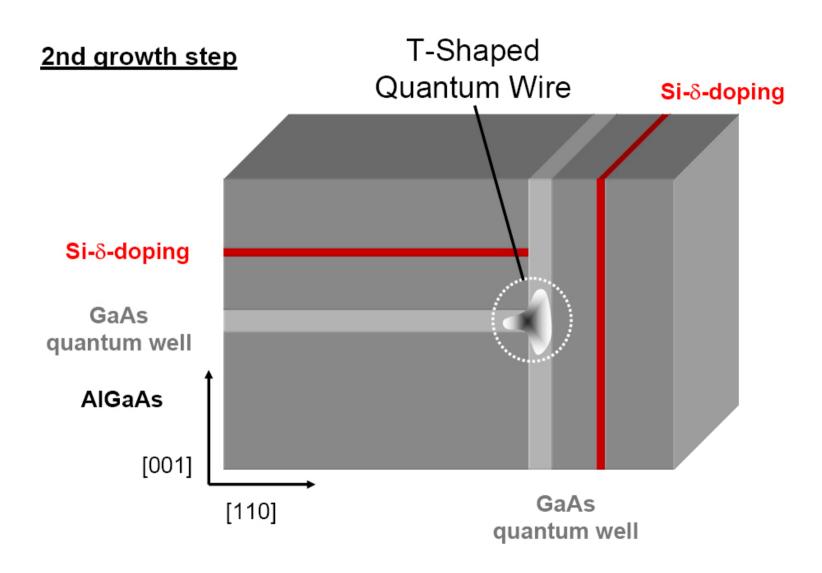






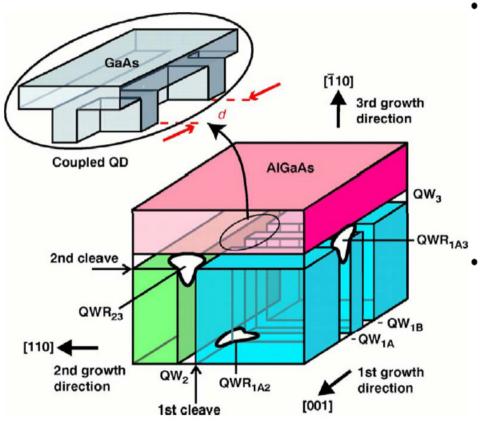












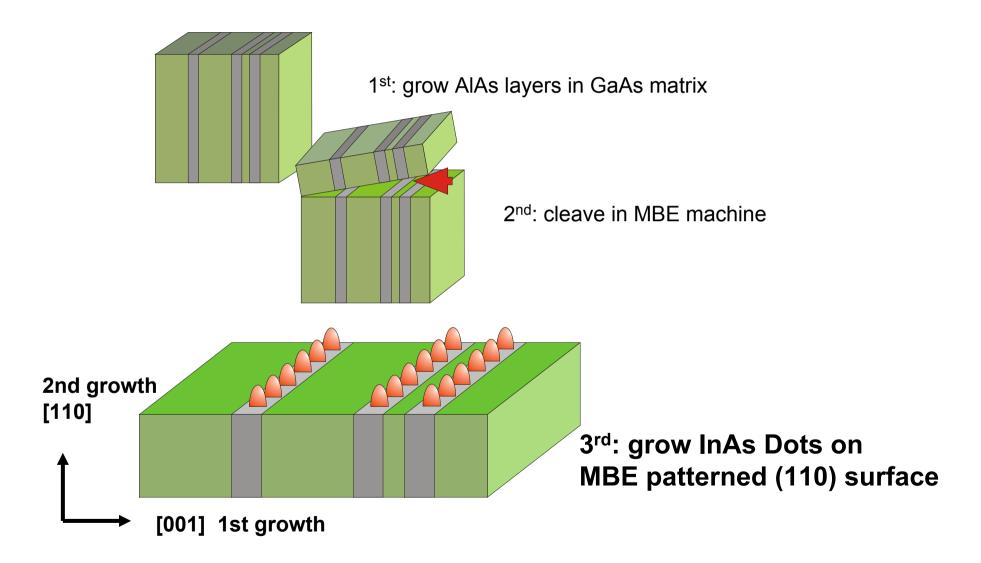
- Two cleaving steps enable fabrication of QDs and artificial molecules
  - 1st growth on (001) GaAs
  - 2nd growth on (011)
  - 3rd growth on  $(01\overline{1})$
  - At intersection between three quantum wells
    - Weaker localisation
    - Lower energy state





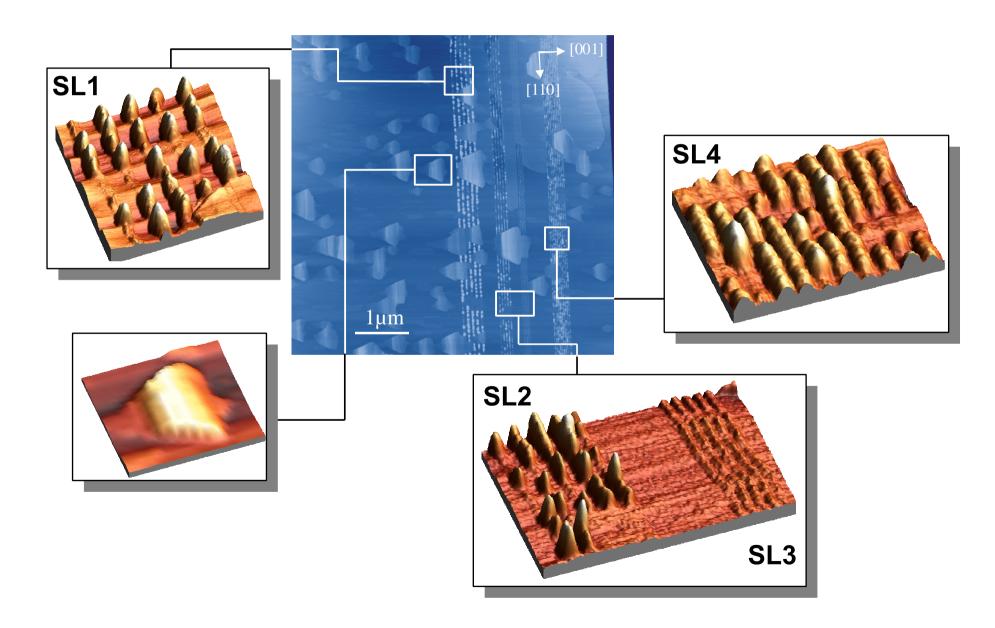






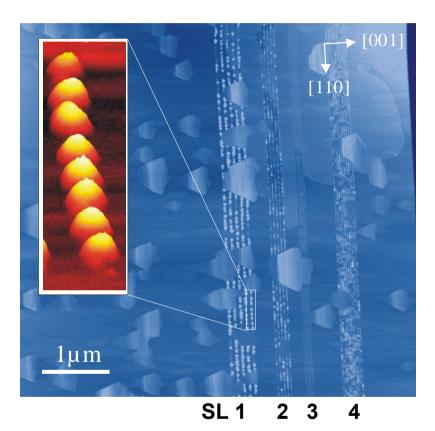












	SL1	SL 2	SL3	SL4
AlAs-width [nm]	32	20	11	20
dot width [nm]	35	22	12	22
dot height [nm]	13 ±4	7 ±1	3	7 ±1
density [dots/µm]	17	14		

quantum dot size correlated with AlAs-width

**♦** create chessboard-structure?





### **Advantages**

- Very high crystal quality
- Confinement for both electrons and holes
- Flexibility

### **Disadvantages**

- Relatively low confinement energies (∆~10meV)
- Complex crystal growth and fabrication
- (011) surface not purely As or Ga terminated like (100)





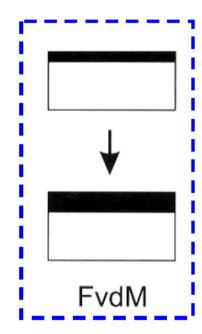
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## **Self Assembling QDs Epitaxial Growth Modes**

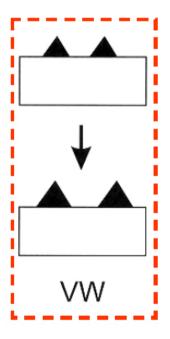


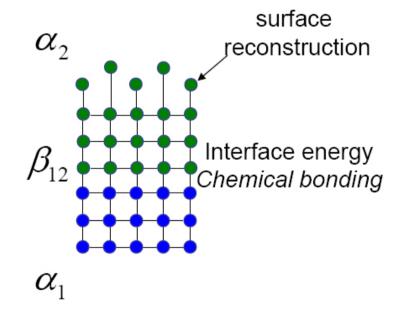
- Equilibrium crystal growth driven by thermodynamic forces
  - Surface  $(\alpha)$  and interface  $(\beta)$  energies
  - Two growth modes = Frank-van der Merwe (FvdM), Volmer-Weber (VW)



$$\alpha_2 + \beta_{12} < \alpha_1$$

Deposited Material Wets Substrate





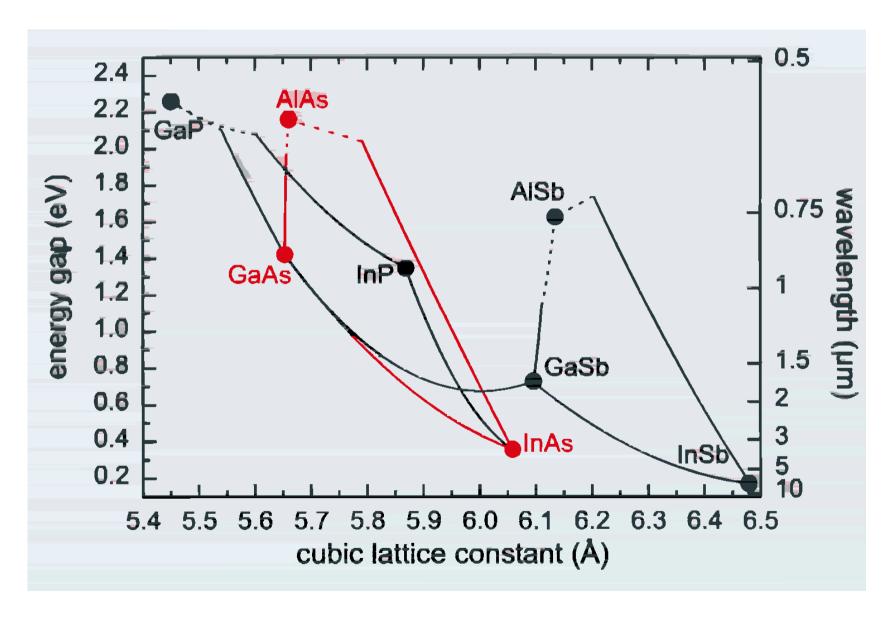
$$\alpha_2 + \beta_{12} > \alpha_1$$

Clustering reduces free energy



## **Self Assembling QDs Energy Gap vs. Lattice Constant**

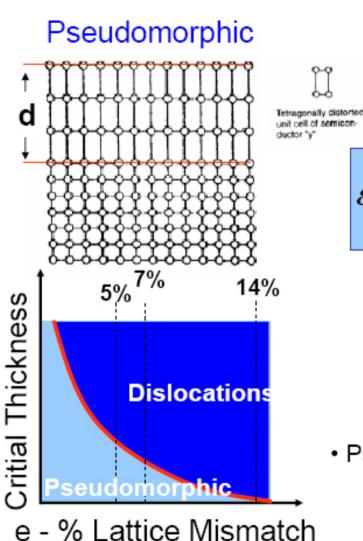




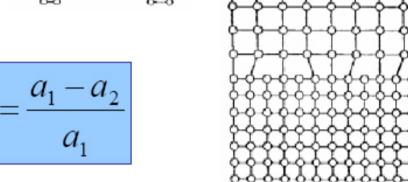


## Self Assembling QDs Strained Layer Epitaxy









Dislocated

For (001) growth Strain Energy

$$E_s \propto \varepsilon^2 d$$

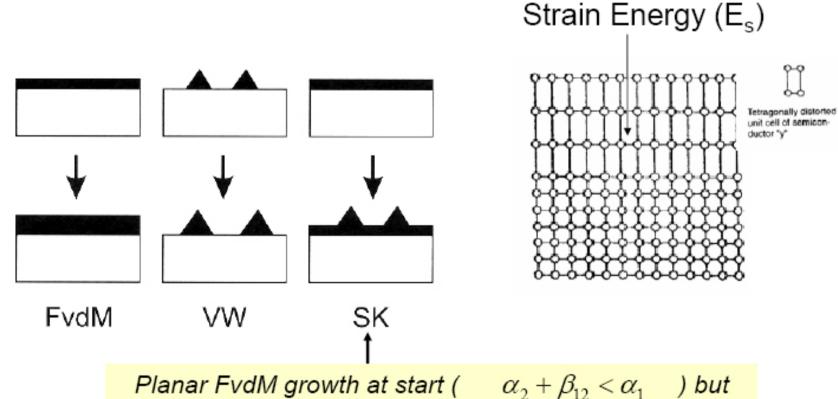
- Pseudomorphic growth
  - Strain energy increases ~ linearly with d



### **Self Assembling QDs Growth Modes in Strained Systems**



 Switching between FvdM and VW growth possible due to increase of strain energy during heteroepitaxy...



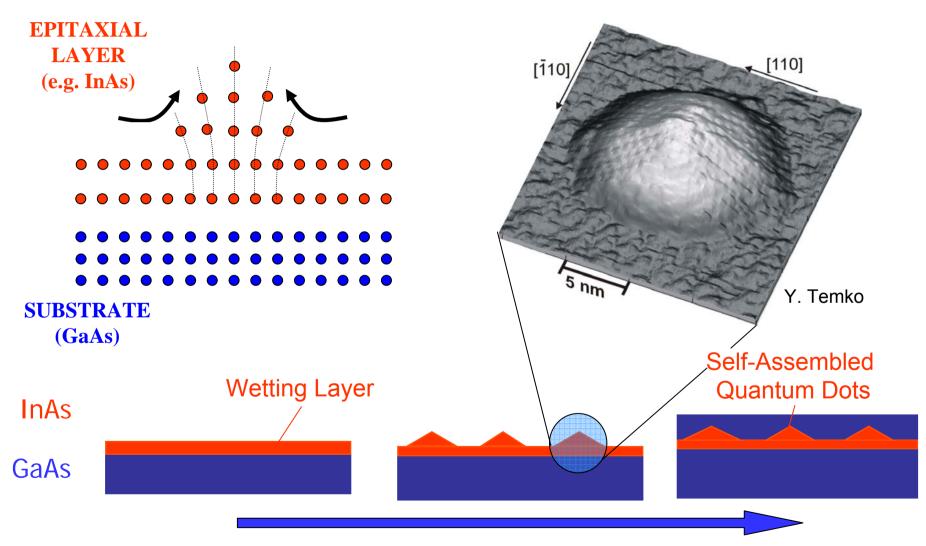
Planar FvdM growth at start (  $\alpha_2 + \beta_{12} < \alpha_1$  ) but growing strain energy ( $E_s$ ) drives a change from FvdM to VW like growth  $\rightarrow$  Stranski-Krastanow Mechanism



### **Self Assembling QDs Stranski-Krastanov Growth**



Nanostructures formed during lattice mismatched epitaxy (e.g. InAs on GaAs)

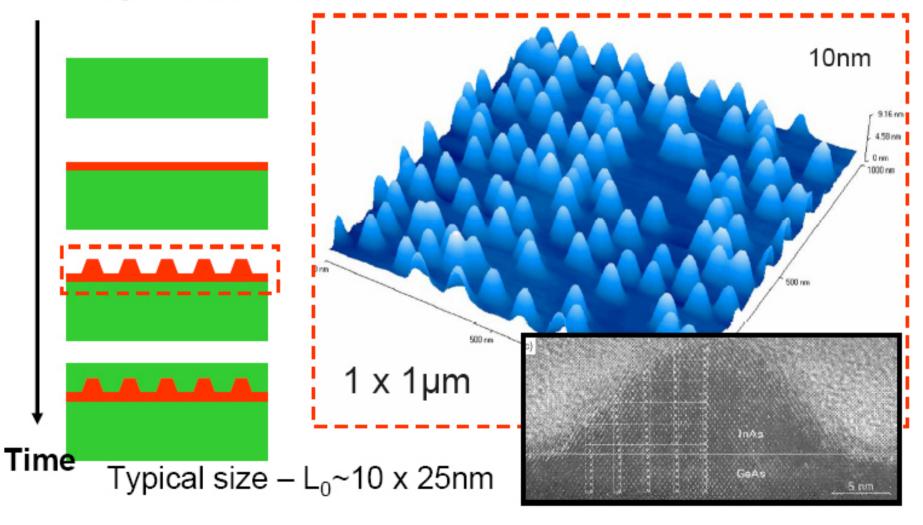




### **Self Assembling QDs Islands** → **Quantum Dots**



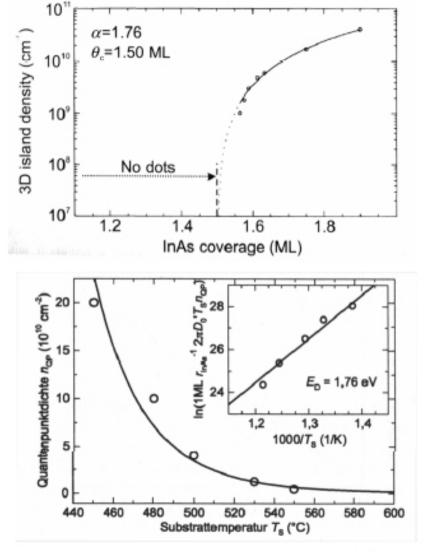
- Formed during Stranski-Krastanow growth of lattice mismatched materials
  - e.g. GaAs substrate + InAs islands + GaAs cap

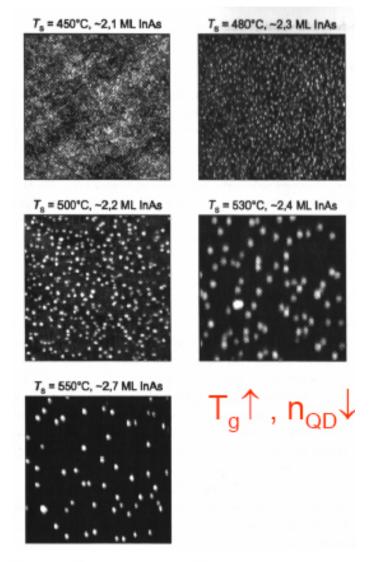




### **Self Assembling QDs Influence of Growth Parameters**







Growth conditions control QD size, density and composition

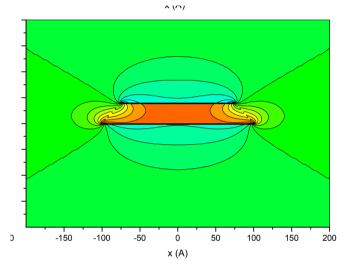


# **Self Assembling QDs Strain**

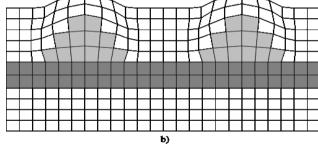


Upper layers of dots tend to nucleate in strain field generated by lower layers

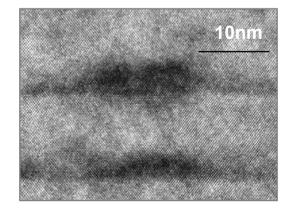
Strain field extends outside buried QD

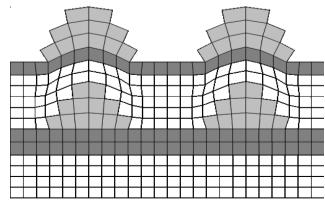


a)



Transmission Electron Micrograph of single coupled QD molecule



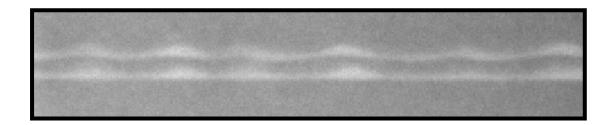


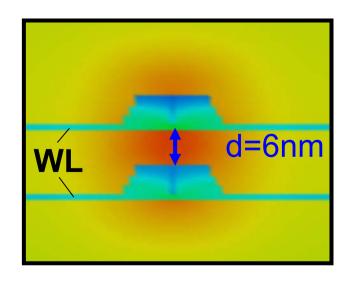


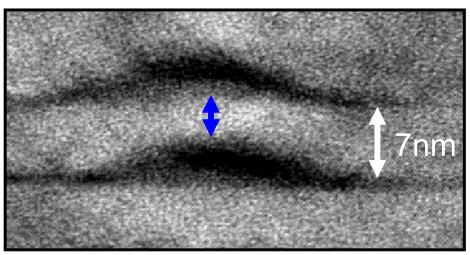
### **Self Assembling QDs Strain**



- InGaAs-GaAs self assembled QD-molecules
- Self alignment via strain field



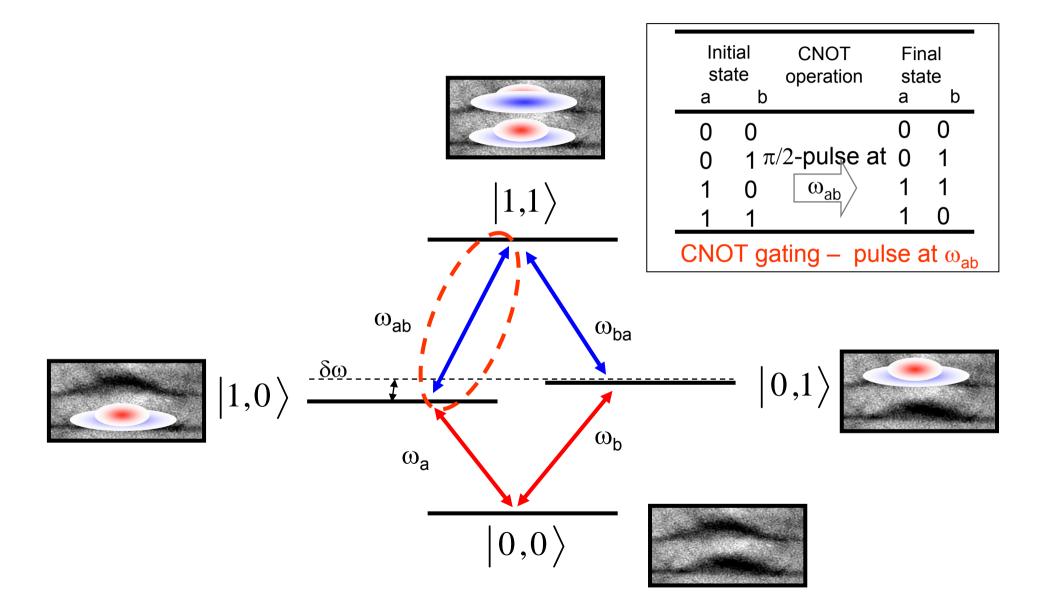






# Self Assembling QDs Quantum Logic







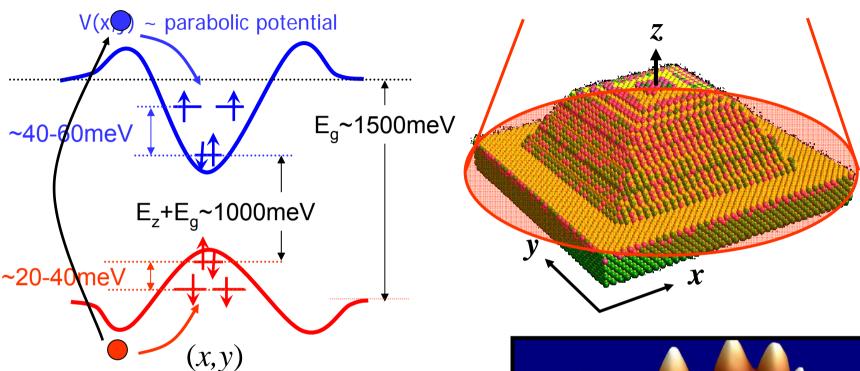


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    - Electronic Structure



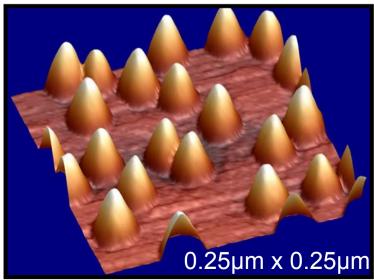
## **Electronic Structure Single Dot Spectroscopy**





### Inhomogeneous broadening

- →Size, shape and composition fluctuations
- →Limits range of physical phenomena investigable

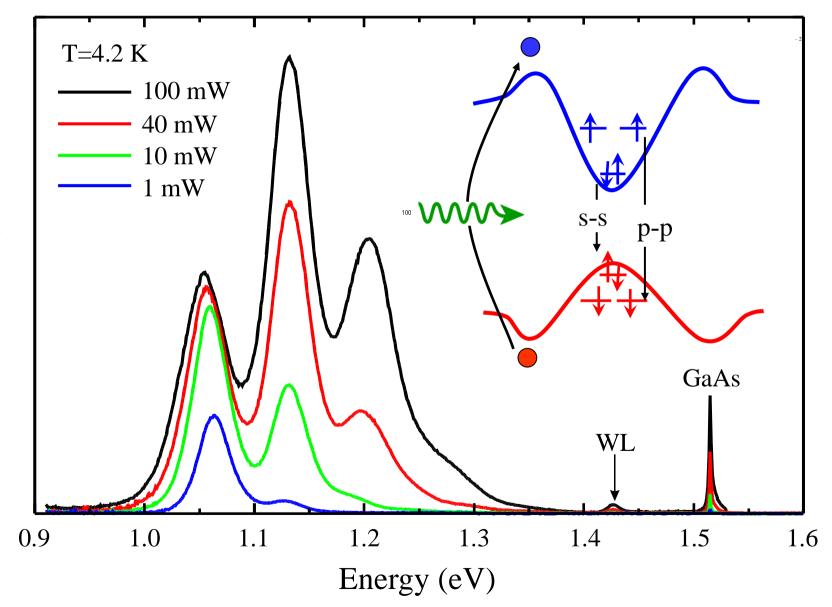




### **Electronic Structure Photoluminescence of QD Ensemble**





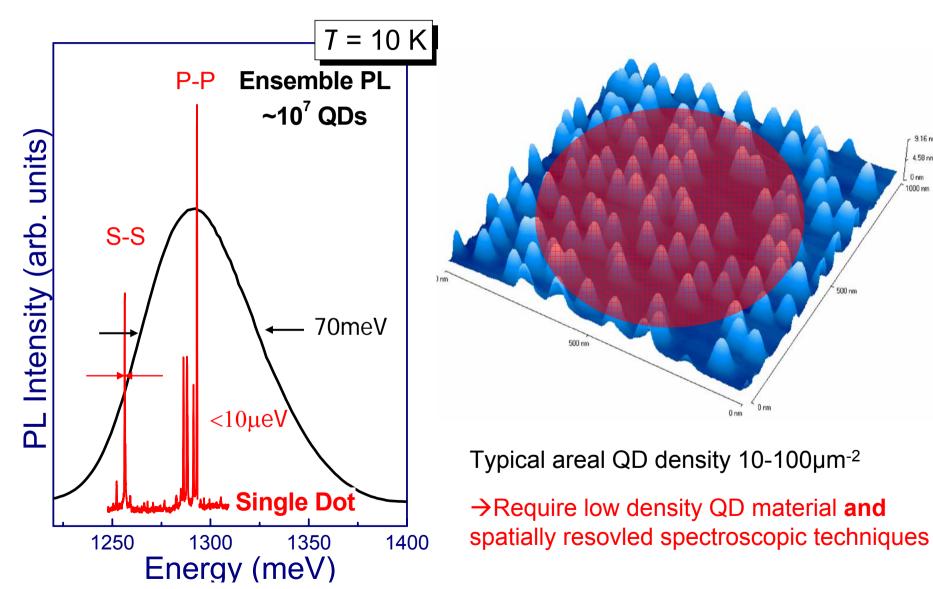




### **Electronic Structure Ensemble** → **Single Quantum Dot**



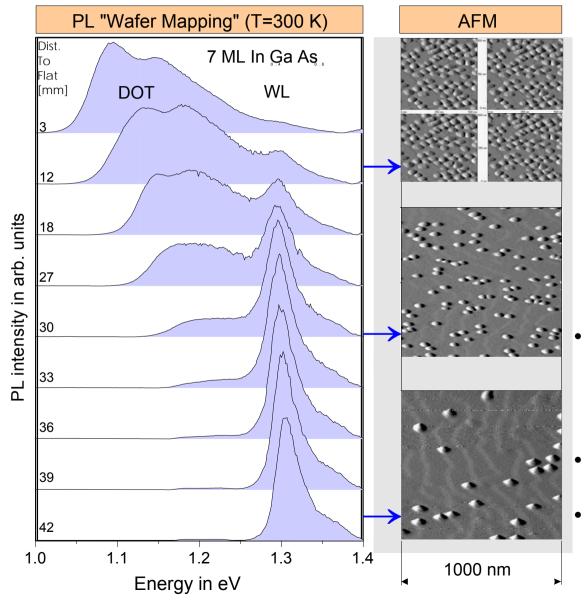
4.58 nm

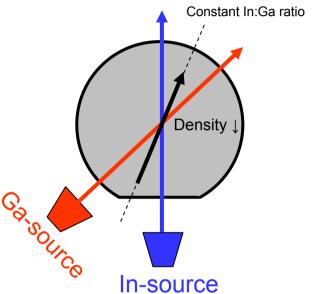




### **Electronic Structure Low Quantum Dot Density**





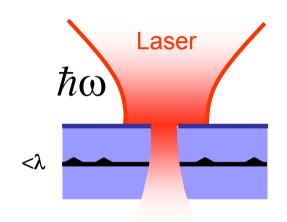


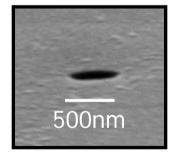
- Growth without substrate rotation
  - Control of In-gradient and In:Ga Ratio
- QD density 300-10μm<sup>-2</sup>
- Characterization using PL and AFM



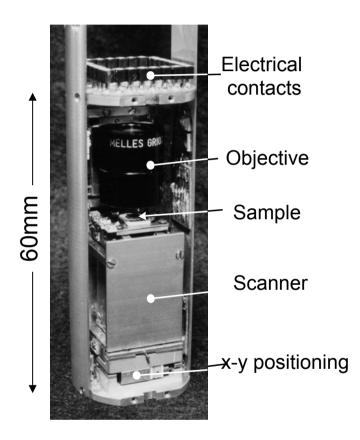
# Electronic Structure Spatially Resolved Spectroscopy



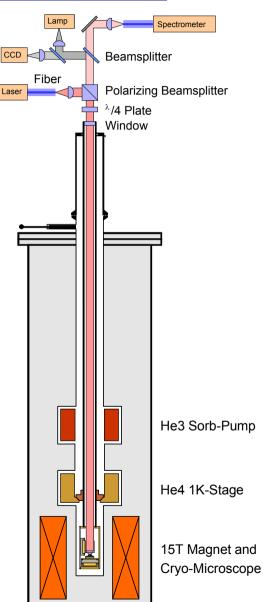




Shadowmask Apertures



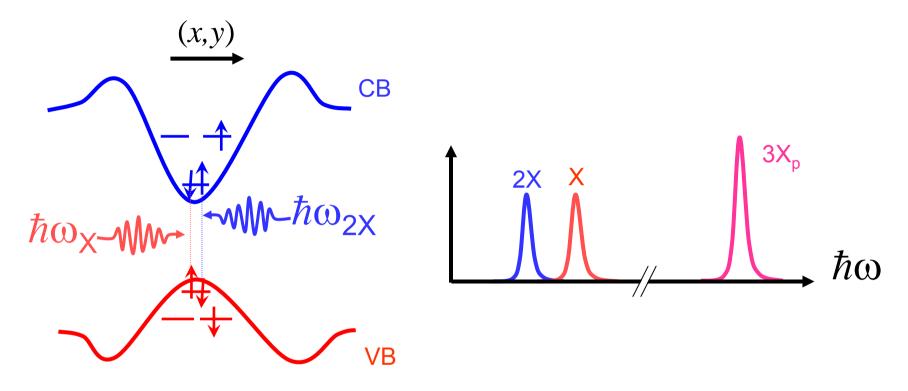
Cryogenic Microscope





## **Electronic Structure Optical Nonlinearities**





Each occupancy state (1e + 1h, 2e + 2h...) has <u>distinct</u> transition frequency

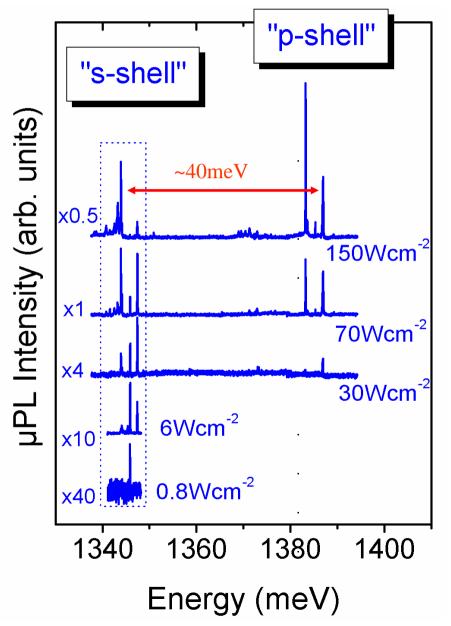
Application of single dots for quantum information science?
Charge and spin qubits...

Deterministic single photon sources...

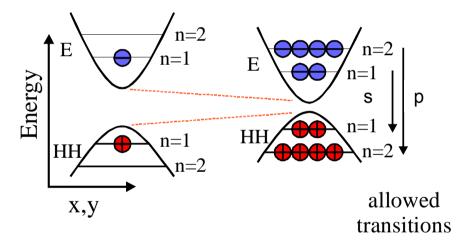


### **Electronic Structure Single QD Photoluminescence**





- Power controls occupancy
  - Low → Single emission line
  - High → Two groups of lines

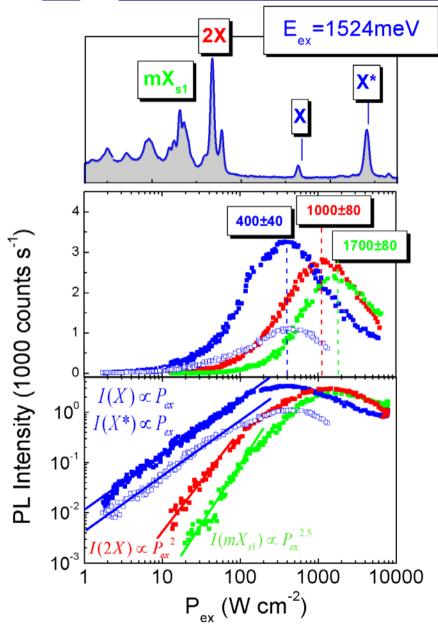


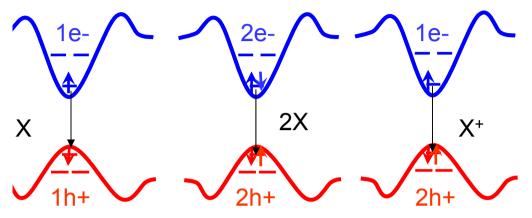
- Two "energy scales"
  - Quantisation energy ~40meV
  - Few particle interactions ~meV



### **Electronic Structure Identification of Occupancy States**







QD occupancy states (X, 2X, 3X...)

→ Identified from power characteristics

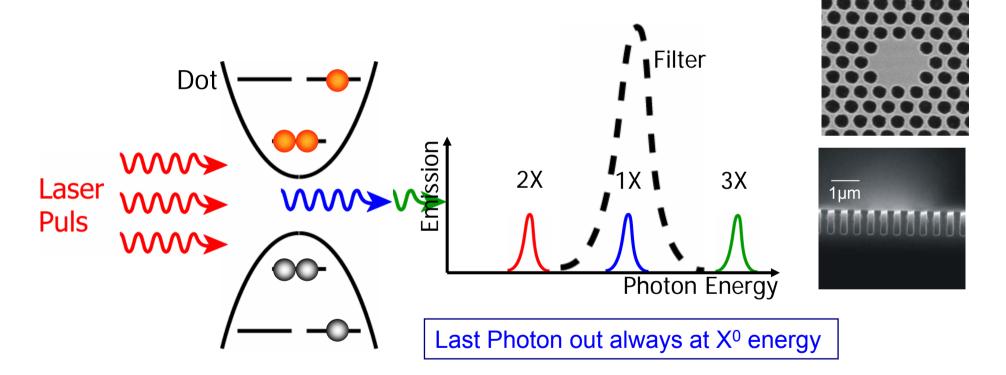
$$P(q) = \frac{N^q \exp(-N)}{q!}$$
 Prob. dot occupied with "q" e-h pairs Seneration rate

→Two "single exciton" lines (X and X\*)





Pulsed optical excitation of a single dot



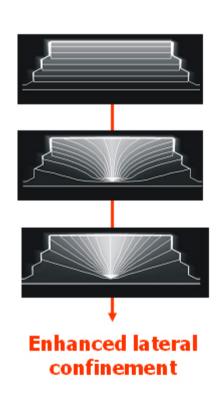
Each external laser pulse produces single photon at X0 energy



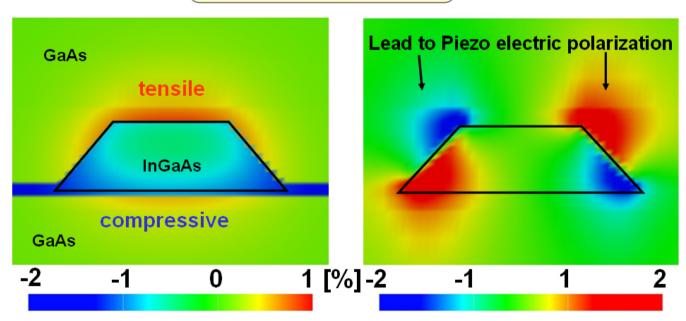
### **Electronic Structure Calculate Strain**



Minimization of elastic energy in continuum model.



$$E_{EL} = \frac{1}{2} \int_{V} C_{ijkl} \, \varepsilon_{ij}(\mathbf{r}) \varepsilon_{kl}(\mathbf{r}) dV$$





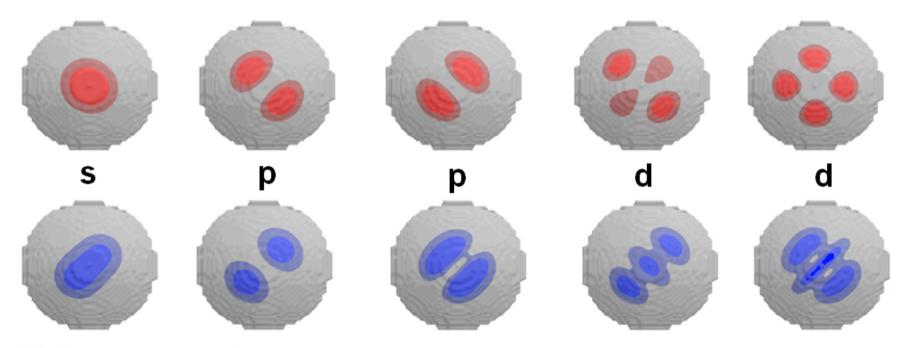
### **Electronic Structure Calculate the Quantum States**



Solve single- or multi-band (k.p) Schrödinger equation

$$\nabla \frac{1}{m_c^*(\mathbf{r})} \nabla \Psi(\mathbf{r}) + E_c(\mathbf{r}) - e\Phi(\mathbf{r}) = E\Psi(\mathbf{r})$$

#### **Electron** wavefunctions



Hole wavefunctions





# Nanoscale islands form during strain driven self-assembly

- Formation is driven by thermodynamic forces
- Size of islands is self-limiting 10-100nm range
- Realised in many materials systems
  - (e.g. InAs on (Al)GaAs, Ge on Si, InAs on InP...)
- Already incorporated into many optoelectronic devices
  - Lasers, LEDs, Detectors, Non-Classical Light emitters, Hardware for quantum computation





### **Advantages**

- Large confinement energies (>60meV)
- High crystal (optical) quality
- High areal density (10<sup>10</sup>-10<sup>11</sup>cm<sup>-2</sup>)
- Weak coupling to their environment
- Multiple layers of dots can be readily fabricated

### **Disadvantages**

Homogeneity - size, shape and morphology fluctuations





- Jon Finley
- Dimitri Kovalev
- Martin Stutzmann
- Andreas Kress, Felix Hofbauer, Michael Kaniber
- WSI (E24)

References: Please ask for special topics.